

Effects of domains in phonon conduction through hybrid boron nitride and graphene sheets

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We theoretically investigate the phonon propagation and thermal conductivity κ in hybrid boron nitride and graphene sheets. By using a real-space Kubo-computational transport scheme, large and disordered graphene structures are simulated, introducing disk-shaped domains with varying sizes of 2 to 8 nm and concentrations ranging from 0% to 100%. A strong influence of the domain size and concentration on the transport properties is obtained. The mean free paths are minimized at 50% domain concentration, and stronger suppression of κ is achieved with smaller domains. It is found to decrease by up to 65% at room temperature when the domain size is 2 nm. These results are beyond the scope of any effective medium approximation.

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I. INTRODUCTION

There is growing interest in two-dimensional (2D) materials for nanoelectronics applications. In this respect, graphene^{1,2} and BN monolayers^{3–5} are sources of exciting new physics.^{6,7} The number of materials that have been exfoliated as single layers keeps increasing as shown very recently by the cases of Bi₂Te₃⁸ and MoS₂.⁹ In parallel with their syntheses, the electronic and thermal conduction properties of these materials are the target of very recent theoretical and experimental investigations, and they continue to yield surprises.^{10–18}

An exciting possibility has been triggered by the demonstration of domain formation in these layers. For example, synthesis of isotopic graphene domains was reported,¹⁹ and hybrid BN-C domains were evidenced.^{20,21} The possible impact of isotopic domains on thermal conduction was suggested in Ref. 22, which predicted a strong size dependence of the thermal conductivity of graphene when small isotopic ¹³C clusters were embedded in the ¹²C graphene host. That study was based on the assumption that each cluster scatters phonons independently and that multiple-scattering effects can be ignored, an approach dating back to the effective-medium virtual crystal approximation (VCA), already employed by Abeles in his analysis of alloy thermal conductivity.²³ This is reasonable for small isotopic concentrations, but it becomes questionable whether such a picture holds for intermediate concentrations far from 0% or 100%. Thus, in order to tackle the problem at any concentration without resorting to any approximations, we have implemented an order- N real-space Kubo formalism,^{24,25} which allows us to simulate large systems.

In this work, we investigate the effects of domains on phonon transport through hybrid BN-graphene sheets as illustrated in Fig. 1. The remainder of this paper is structured as follows. In Sec. II, we briefly introduce the real-space Kubo method. In Sec. III, we investigate the effects of the BN concentration, size of the domains, and B isotopes on thermal conductivity. Finally, we present our conclusions in Sec. IV.

II. METHODOLOGY

In the real-space Kubo approach,^{24,25} we calculate the time-dependent diffusivity $\mathcal{D}(\omega, t) = \chi^2(\omega, t)/t$, where χ^2 is the mean-square spreading of the wave packets of frequency ω along the transport direction x . \mathcal{D} behaves as $v_x^2 t$ for short times, where v_x is the average velocity along the x direction. It saturates to $\mathcal{D}_{\max} \approx 2v_x^2 \tau_{\text{tr}}$ in the diffusive regime, where τ_{tr} is the average transport time, and then decreases due to localization effects. Considering $v_x^2 = v^2/2$ in two dimensional systems, the mean free path (MFP) can be obtained via $\ell(\omega) = \mathcal{D}_{\max}(\omega)/\sqrt{2}v_x(\omega)$. Mean-square displacement can be expressed as

$$\chi^2(\omega, t) = \frac{\text{Tr}\{[X, U(t)]^\dagger \delta(\omega^2 - D)[X, U(t)]\}}{\text{Tr}[\delta(\omega^2 - D)]}, \quad (1)$$

where X is the position matrix, D is the mass-normalized dynamical matrix, and $U(t) = e^{-i\sqrt{D}t}$ serves as the evolution operator. The trace in Eq. (1) can be efficiently calculated through an average over a few initial random-phase states and can be further evaluated by using a Chebyshev expansion of $U(t)$ and the Lanczos method. It yields a mode-averaged value for the mean-square displacement and the mean free path. Possible shortcomings due to averaging over decoupled modes are avoided as explained below.

The fourth-nearest-neighbor cutoff is proven to describe phonons of graphene accurately,²² and the fourth-nearest-neighbor force-constant (4NNFC) scheme allows a fairly accurate description of phonon spectra and heat transport properties of sp^2 -hybrid carbon systems.^{26,27} We use the 4NNFC parameters from the pure graphene system²⁷ and treat B and N atoms in terms of their mass differences as an approximation. In Fig. 2, phonon dispersions of single-layer BN obtained with the 4NNFC-mass-difference approximation are compared with those obtained from Tersoff-type parametrizations,²⁸ density functional theory calculations,²⁸ and experiments on bulk hexagonal BN.²⁹ It is observed that DFT calculations

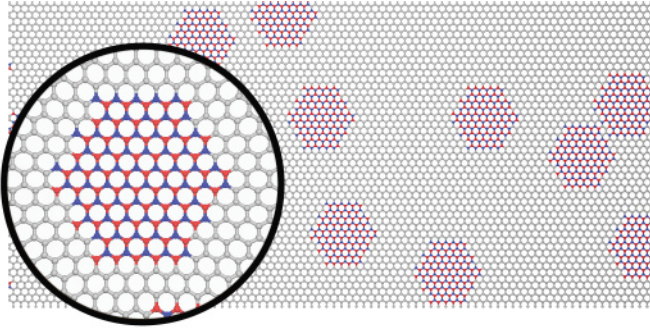


FIG. 1. (Color online) A schematic representation of the atomic structure of two-dimensional graphene-BN heterostructures. Here, BN clusters with 2-nm diameters are embedded inside the graphene.

yield the best agreement with the experiments. LA and TA modes of the 4NNFC approach are in very good agreement with the DFT results and the experiments. The ZO mode obtained from the 4NNFC approach agrees fairly well with the DFT results, and it gives a better estimation than the Tersoff parametrization. On the other hand, the 4NNFC approach cannot predict the ZA mode as well as it does the LA, TA, and ZO modes. It overestimates the group velocities of ZA phonons and underestimates the out-of-plane energy gap, namely, the ZA-ZO splitting around K . The 4NNFC approach underestimates the group velocities of the LO and TO modes, but these high-energy modes have only a minor contribution to the room-temperature thermal conductivity.

Comparing the 4NNFC approach with the Tersoff parametrization, we find that the LA and TA modes are equally well described by both methods. The ZO mode is better described by the 4NNFC approach while the Tersoff parametrization more accurately predicts the ZA mode. While both the 4NNFC and Tersoff approaches underestimate the ZA-ZO splitting around K , the 4NNFC method gives a better estimation of the group velocities of these phonons. The maximum and minimum frequencies of the in-plane optical

modes (LO and TO) are estimated correctly with the 4NNFC approach while the Tersoff parametrization overestimates them.

In summary, the in-plane modes LA, TA, TO, and LO are well described by the 4NNFC mass-difference approximation. Considering the phonon populations at room temperature, we conclude that their contribution to thermal conductivity is accurately estimated by the 4NNFC approach. The out-of-plane ZA and ZO modes are not predicted as well as the in-plane modes. Experimental results and DFT calculations show that the out-of-plane modes display a gap in the 350–600 cm^{-1} frequency window. In parallel to this, we estimate that the shortest mean free paths in order of 10 nm or below lie within the 450–600 cm^{-1} region. Since the simulation window's size is $1 \mu\text{m} \times 1 \mu\text{m}$, we predict a transport gap at these energies. Therefore, we conclude that the 4NNFC mass-difference approach reproduces thermal transport properties of hybrid BN-graphene heterostructures satisfactorily.^{28–30}

The quadratic out-of-plane vibrational modes are completely decoupled from the in-plane modes,²⁷ so we can study the MFPs separately, simply by limiting the initial random-phase state to out-of-plane vibrations or in-plane vibrations. This also enables us to distinguish the fundamental differences between in-plane and out-of-plane vibrations. The system size is chosen to be 100 nm \times 100 nm, and periodic boundary conditions are employed. It is known that the MFP diverges as $\omega \rightarrow 0$. Ideally, one needs to consider an infinite system to be able to calculate MFPs for the entire spectrum, which is practically impossible. On the other hand, below a frequency threshold the MFP becomes comparable with that of our assumed sample size ($1 \mu\text{m} \times 1 \mu\text{m}$), so the effect of domains at these frequencies is masked by the sample boundary scattering. Accordingly, we calculate the MFP down to 50 cm^{-1} , and intrinsic scattering is neglected below this frequency.

III. RESULTS

We investigate the effect of the overall concentration of BN as well as the influence of the domain size on the mean free path in the cases with pure and isotopic mixtures of boron. In our simulations the domains are disk-shaped, and their diameters are set to be 2, 4, or 8 nm (Fig. 1).

A. Concentration effect

In studying the effects of concentration, domain diameter is fixed to be 2 nm with a single isotopic species of boron, namely ^{11}B . We consider five different atomic concentrations of BN, 20%, 40%, 50%, 60%, and 80%, while the ratio of B and N is 1:1 in all cases. For the concentrations 20% and 40%, the host matrix of the sheet is graphene while BN is considered to be the host matrix for the 60% and 80% concentrations. For the 50% concentration, both of the matrices are considered. Our calculations show that different choices of the host give similar results owing to disorder average.

The calculated transport MFPs are plotted in Fig. 3. The density of states (DOS) is very low around 537 cm^{-1} for graphene, corresponding to the intersection of the dispersions of the ZA and ZO modes at the K point of the Brillouin

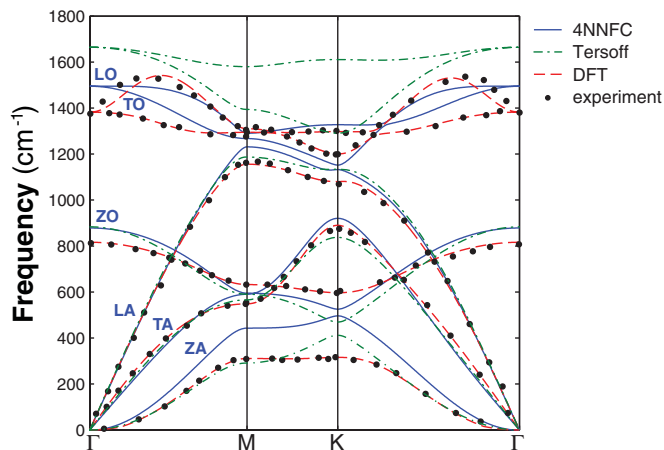


FIG. 2. (Color online) Comparison of the phonon dispersions of a single sheet of BN obtained from the fourth-nearest-neighbor force-constant model with Tersoff-type parametrization (data taken from Ref. 28) and density-functional-theory calculations. Experimental results are also included for comparison.

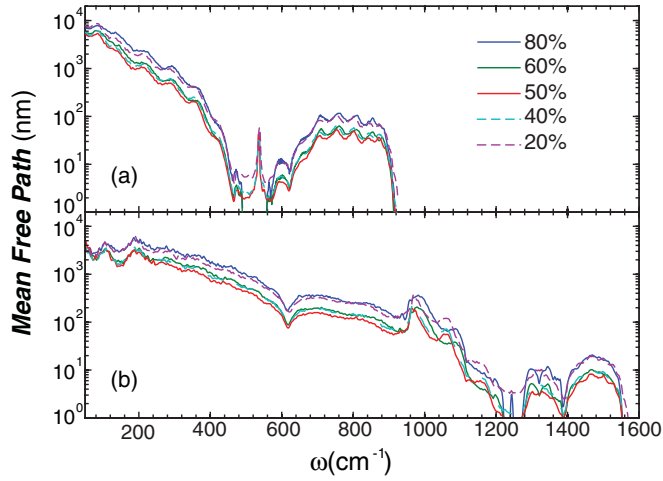


FIG. 3. (Color online) Mean free paths for different concentrations at a fixed domain size of 2 nm are plotted for (a) out-of-plane and (b) in-plane modes, separately.

zone. In the case of a pure BN sheet, these modes do not intersect but repel each other due to broken-sublattice symmetry, and the DOS is even lower than that of graphene.³¹ As a result, out-of-plane modes are suppressed very strongly at these energies as the BN concentration is increased. In order to gain more insight, we also performed calculations with random distributions of individual C atoms in BN hosts (Fig. 4). In this case, the MFPs decrease with increasing C concentrations for both in-plane and out-of-plane modes, the minimum value being achieved at a 50% concentration. We note that at lower frequencies, out-of-plane modes have shorter MFPs for the atomic distributions while for in-plane modes, cluster distributions have shorter MFPs below 100 cm^{-1} . An interesting feature of the cluster distributions becomes more apparent upon comparison with the atomic distributions, namely, the oscillations in the MFP as a function of frequency are characteristic of the clustering. VCA predicts a quadratic dependence of the MFP on the density within the mass-difference approximation $\ell^{-1} \sim d(1-d)$. Our results are in

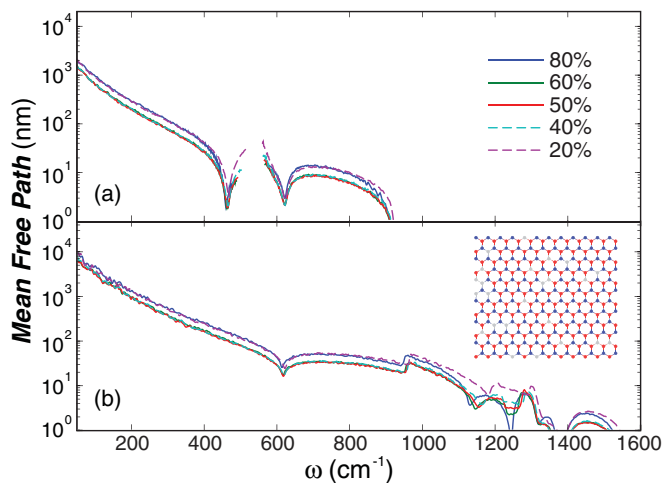


FIG. 4. (Color online) C atoms randomly distributed in the BN host (see the illustration in the inset). MFPs with different concentrations of C atoms for (a) out-of-plane and (b) in-plane modes.

agreement with the prediction that the minimum MFP is achieved at $d = 0.5$, but simple quadratic density dependence is not reproduced by our calculations in which $\lambda_{d=0.5}/\lambda_{d \neq 0.5}$ is generally smaller than that of the VCA prediction as expected due to the incorporation of multiple-scattering effects in our calculations.

B. Domain-size effect

In order to investigate the influence of the domain sizes, we keep the concentration fixed at 50% and generate random samples with domains having diameters $d = 2, 4$, or 8 nm. The host is graphene, and all B atoms are of the same isotopic species, ^{11}B . We observe a general increase in the MFPs with increasing domain size (Fig. 5). In the long wave limit ($\omega \rightarrow 0$), MFPs should decrease with increasing size;^{22,32} therefore, at very low temperatures, thermal conductivity can be reduced by increasing the domain size. However, the MFPs at these low frequencies become comparable to our chosen sample size of 1 μm . We also compare the MFPs due to disk-shaped domains with that due to hexagonal scatterers consisting of only 6 atoms where the diameter is $\simeq 0.3$ nm. For $\omega > 200 \text{ cm}^{-1}$, MFPs due to hexagonal scatterers are generally smaller. At $\omega < 200 \text{ cm}^{-1}$, the crossover takes place for the in-plane modes, and long-wavelength phonons are scattered less effectively by the small-diameter hexagonal domains. At the other extreme for $\omega > 1300 \text{ cm}^{-1}$, hexagons give rise to very short localization lengths due to vanishingly small MFPs. The domains considered here are considerably larger than those in Ref. 22. For these large sizes and frequencies down to 200 cm^{-1} , scattering is in the short-wave regime, characterized by MFPs proportional to the domain diameters. This is evident from the overlapping curves of the plots of ℓ/d ($d = 2, 4, 8$ nm). In contrast, in the long-wavelength, or Rayleigh, regime, the MFP for a fixed-domain surface fraction in two dimensions would be inversely proportional to the square of the domain size. Rayleigh theory predicts the cross section of a cluster to be proportional to the square of the number

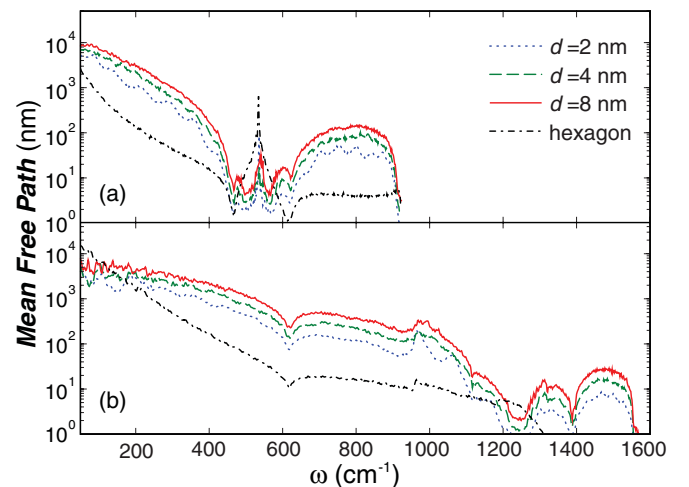


FIG. 5. (Color online) MFPs for different domain sizes at a fixed concentration of 50% for (a) out-of-plane and (b) in-plane modes. MFPs with domains consisting of single hexagons are also plotted for comparison.

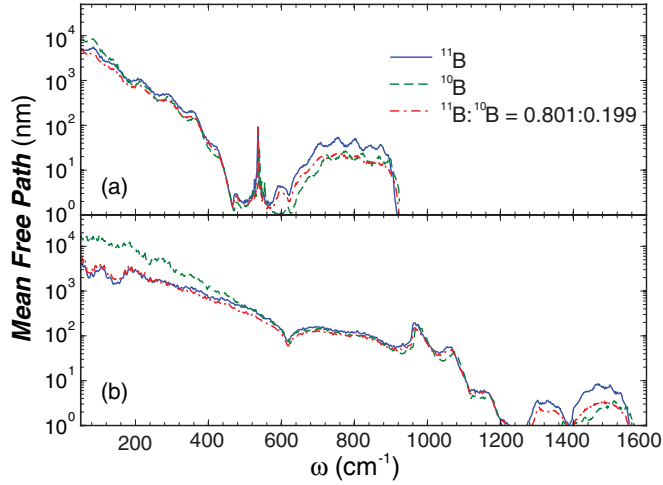


FIG. 6. (Color online) MFPs for different B isotopic compositions at a 50% concentration and 2-nm domain size. (a) Out-of-plane and (b) in-plane modes.

of atoms in the cluster at low frequencies. Our calculations confirm that scattering per atom is stronger for large clusters at low frequencies for which the crossover frequency is around 200 cm^{-1} . The MFPs of quadratic out-of-plane modes show an $\omega^{-3/2}$ dependence at low frequencies as predicted by Rayleigh scattering theory²² while those of linear in-plane modes display an ω^{-3} dependence.

C. Isotope effect

We now study the effect of isotopic distribution on a graphene host with a 50% BN concentration, keeping the domain size fixed as $d = 2 \text{ nm}$. Boron has two stable isotopes, ^{11}B and ^{10}B with natural abundances of 80.1% and 19.9%, respectively. Three different B compositions are considered: pure ^{11}B , pure ^{10}B , and a mixture of the two with their natural abundances. For isotopically pure ^{10}B , the average mass of a BN cluster is approximately equal to that of graphene while it is larger for pure ^{11}B . As shown in Fig. 6, MFPs for the pure ^{10}B case are longer than those for the pure ^{11}B case at low frequencies. Though the mass difference between ^{10}B and ^{12}C is larger than the difference between ^{11}B and ^{12}C , the negative mass difference is canceled out by the positive mass difference between ^{14}N and ^{12}C , leading to stronger scattering by ^{11}B than by ^{10}B at low frequencies. In other words, low-frequency modes are more sensitive to the average mass of the BN clusters while at high frequencies, the deviation of atomic masses from the average are more effective. One also observes that the curves for the mixture case do not simply fall in between the curves of the isotopically pure cases. At low frequencies, they follow the ^{11}B curve, which is more abundant, but at higher frequencies, this tendency is lost.

D. Thermal conductivity

In a sample of length L with perfect contacts, κ can be approximated as³³

$$\kappa_{o/i} = \frac{1}{h} \int_0^\infty \frac{d\omega}{2\pi} \hbar \omega \frac{\partial f}{\partial T} \left(1 + \frac{4}{\pi} \frac{L}{2\ell_{o/i}(\omega)} \right)^{-1} \mathcal{T}_{o/i}(\omega), \quad (2)$$

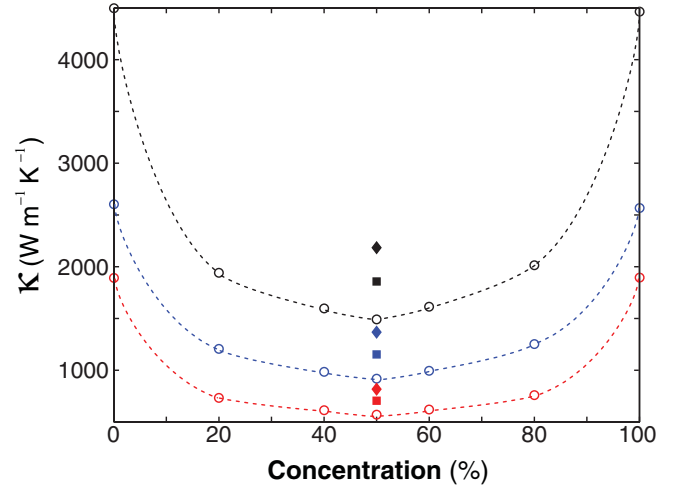


FIG. 7. (Color online) Room-temperature thermal conductivity for a 2-nm domain size versus the concentration of BN contributed by out-of-plane modes (blue/dark gray), in-plane modes (red/light gray), and all modes (black). Dashed curves are plotted as guides to the eye. At 50% concentration, two additional domain sizes are considered. Circles: 2 nm; squares: 4 nm; and diamonds: 8 nm.

assuming that the phonon dispersions and mean free paths are isotropic. This approximation becomes more accurate as the scattering rates are increased and the modes are mixed more effectively. In Eq. (2), i and o denote in-plane and out-of-plane modes, respectively. $h = 3.35 \text{ Å}$ is the graphitic interlayer spacing, f is the Bose function, and T is the ballistic transmission, which is computed from the phonon dispersions in the entire Brillouin zone. For $\omega < 50 \text{ cm}^{-1}$, phonons are assumed to propagate ballistically, and the term in the brace is approximated by 1 (see Sec. II). The calculated conductivities for a 2-nm domain size for different concentrations are plotted in Fig. 7. Conductivities for $d = 4$ and 8 nm are also indicated for a 50% concentration. Conductivity changes rapidly near 0% and 100% concentrations, but this change is considerably slower around a 50% concentration. The conductivity is nearly symmetric around a 50% concentration with a minimum where it is reduced by 65%, compared to pure graphene. Such symmetric behavior could be altered with a more accurate description of the out-of-plane vibrations, but we do not expect a major change. A considerable increase in the conductivity can be observed when the domain size increases. A cautionary note is in order here. Since we do not include the effect of anharmonicity, the conductivity of pure graphene depends linearly on L . In contrast, the conductivities of the hybrid-domain systems are nearly independent of length. Therefore, the obtained reduction in the thermal conductivity also depends on the sample's length.

IV. CONCLUSIONS

Using a real-space method to compute phonon propagation, we have reported the strong effects of domains on thermal conductivity and have shown that hybrid BN-C layers have interesting possibilities regarding the tunability of their thermal conductivities by simultaneously adjusting the domain size and concentration. Suppression of the thermal

conductivity is less pronounced with larger domains while a 65% reduction in room-temperature thermal conductivity is possible when the clusters have 2-nm diameters and a 50% concentration.

Note added in proof. Another set of parameters has been recently derived for hexagonal BN interatomic potentials and applied to study its thermal conductivity in Ref. 34.

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¹K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva, and A. A. Firsov, *Science* **306**, 666 (2004).

²A. Cresti, N. Nemec, B. Biel, G. Niebler, F. Triozon, G. Cuniberti, and S. Roche, *Nano Res.* **1**, 361 (2008).

³K. S. Novoselov, D. Jiang, F. Schedin, T. J. Booth, V. V. Khotkevich, S. V. Morozov, and A. K. Geim, *Proc. Natl. Acad. Sci. USA* **102**, 10451 (2005).

⁴A. Nagashima, N. Tejima, Y. Gamou, T. Kawai, and C. Oshima, *Phys. Rev. Lett.* **75**, 3918 (1995).

⁵D. Pacile, J. C. Meyer, C. O. Girit, and A. Zettl, *App. Phys. Lett.* **92**, 133107 (2008).

⁶A. Ramasubramaniam and D. Naveh, *Phys. Rev. B* **84**, 075405 (2011).

⁷A. Rubio, *Nat. Mater.* **9**, 379 (2010).

⁸D. Teweldebrhan, V. Goyal, M. Rahman, and A. A. Balandin, *App. Phys. Lett.* **96**, 053107 (2010).

⁹B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti, and A. Kis, *Nat. Nanotechnol.* **6**, 147 (2011).

¹⁰A. A. Balandin, S. Ghosh, W. Bao, I. Calizo, D. Teweldebrhan, F. Miao, and C. N. Lau, *Nano Lett.* **8**, 902 (2008).

¹¹J. H. Seol, I. Jo, A. L. Moore, L. Lindsay, Z. H. Aitken, M. T. Pettes, X. Li, Z. Yao, R. Huang, D. Broido, N. Mingo, R. S. Ruoff, and L. Shi, *Science* **328**, 213 (2010).

¹²S. Ghosh, W. Bao, D. L. Nika, S. Subrina, E. P. Pokatilov, C. N. Lau, and A. A. Balandin, *Nat. Mater.* **9**, 555 (2010).

¹³C. Faugeras, B. Faugeras, M. Orlita, M. Potemski, R. R. Nair, and A. K. Geim, *ACS Nano* **4**, 1889 (2010).

¹⁴H. Sevinçli and G. Cuniberti, *Phys. Rev. B* **81**, 113401 (2010).

¹⁵J. Haskins, A. Kinaci, C. Sevik, H. Sevinçli, G. Cuniberti, and T. Çağın, *ACS Nano* **5**, 3779 (2011).

¹⁶A. A. Balandin, *Nat. Mater.* **10**, 569 (2011).

¹⁷J.-W. Jiang, J.-S. Wang, and B.-S. Wang, *App. Phys. Lett.* **99**, 043109 (2011).

¹⁸C. Sevik, H. Sevinçli, G. Cuniberti, and T. Çağın, *Nano Lett.* **11**, 4971 (2011).

¹⁹X. Li, W. Cai, L. Colombo, and R. S. Ruoff, *Nano Lett.* **9**, 4268 (2009).

²⁰L. Ci, L. Song, C. Jin, D. Jariwala, D. Wu, Y. Li, A. Srivastava, Z. F. Wang, K. Storr, L. Balicas, F. Liu, and P. M. Ajayan, *Nat. Mater.* **9**, 430 (2010).

²¹L. Song, L. Balicas, D. J. Mowbray, R. B. Capaz, K. Storr, L. Ci, D. Jariwala, S. Kurth, S. G. Louie, A. Rubio, and P. M. Ajayan, e-print arXiv:1105.1876v1 (unpublished).

²²N. Mingo, K. Esfarjani, D. A. Broido, and D. A. Stewart, *Phys. Rev. B* **81**, 045408 (2010).

²³B. Abeles, *Phys. Rev.* **131**, 1906 (1963).

²⁴W. Li, H. Sevinçli, G. Cuniberti, and S. Roche, *Phys. Rev. B* **82**, 041410 (2010).

²⁵W. Li, H. Sevinçli, S. Roche, and G. Cuniberti, *Phys. Rev. B* **83**, 155416 (2011).

²⁶R. Saito, G. Dresselhaus, and M. S. Dresselhaus, *Physical Properties of Carbon Nanotubes* (Imperial College Press, London, 1998).

²⁷J. Zimmermann, P. Pavone, and G. Cuniberti, *Phys. Rev. B* **78**, 045410 (2008).

²⁸C. Sevik, A. Kinaci, J. B. Haskins, and T. Çağın, *Phys. Rev. B* **84**, 085409 (2011).

²⁹J. Serrano, A. Bosak, R. Arenal, M. Krisch, K. Watanabe, T. Taniguchi, H. Kanda, A. Rubio, and L. Wirtz, *Phys. Rev. Lett.* **98**, 095503 (2007).

³⁰V. N. Popov, *Phys. Rev. B* **67**, 085408 (2003).

³¹L. Wirtz, A. Rubio, R. A. de la Concha, and A. Loiseau, *Phys. Rev. B* **68**, 045425 (2003).

³²N. Mingo, D. Hauser, N. P. Kobayashi, M. Plissonnier, and A. Shakouri, *Nano Lett.* **9**, 711 (2009).

³³N. Mingo and D. A. Broido, *Nano Lett.* **5**, 1221 (2005).

³⁴L. Lindsay and D. A. Broido, *Phys. Rev. B* **84**, 155421 (2011).